Fiz. Metal. i Metalloved. Akad. Nauk SSSR Ural. Filial 3, 309-13 (1996) Sameonov, G. V. Pa-1926

PHYSIOCOCHEMICAL PROPERTIES OF BORON-CARBON ALLOYS

The molten system of boron and carbon has been studied several times previously in particular in relation to its extreme hardness and correction resistance. In 1974, Ridgway' has carried out the x-ray analysis of beron carbide powders (B_0 C). He also determined multing points, specific resistance, coefficient of linear expansion as well as a series of hardness tests characteristic to this compound.

Systematic analysis of the phase diagram for the boron carbide system was started by Graont² in 1939 and was further reported in a later wart². In this latter analysis the region in a phase diagram in the limits of carbon content from 10-505 by weight was determined. It was also found (carbolished) that the boron carbide is formed by the reversible reaction; $(3 + \{1\})$ and the presence of sutectic composition at (30.35 G) was shown. It was supposed that the above sutectic was between boron carbide and carbon phases. G. C. Zhdanov², et al have studied the structure of boron carbide and have established that it consists of a rhombohedral crystalline call with a = 5.60 and c = 12.1 Å*.

In 1955, Ebdanov, et al bave proposed a hypothesis in which they postulated that substances with covalent bonds may form solid solutions under conditions of preserving the directional character of the covalent bonds. In particular, in case of boron carbide, some replacement of the portion of carbon atoms in the (1 b) position of boron atoms which, possessing sp-alestrons, may form a linear valence configuration similar to carbon atoms.

Detailed analysis of this questions has shown that such displacement of earlies atoms in $B_{18}G_{0}$ ($*B_{4}G$) with the formation of $B_{18}G_{0}$ actually takes place; it was also possible to determine the crystal call distances in the $B_{4}G$ crystal with greater precision. The following values were obtained: a = 4.495+0.002 A*, $a = 12.12 \pm 0.002$ A*. Certain physical and chemical characteristics of the $B_{18}G_{0}$ carbide were obtained.

Claser, et al T have carried out as x-ray analysis as well as determination of specific gravity and resistance of the boron-carbon make in the region from 4-60 male \$ carbon. These authors show that the interestance distances in the crystalline phases as well as the specific gravities of these solid solutions continuously increases while the electrical resistance in these solid solutions decreases in the region of 4-26 male \$ carbon. This observation allowed the authors to postulate a continuous series of solid selutions of carbon in boron in the region of concentration that were studied.

Pinelly, Allem⁶ has studied the interatomic distances and specific gravities of malts that may be described by empirical formulas R₂C, R_{3-co}C, B₇C and has shown the possibility of displacement of earlies atoms by berea stems in the solid solution as well as formation of excess bores and earlies malten solutions by the process of infiltration into the existing berea-earlies exystalline shell. The divergence of the available data on the properties of solid solutions of boron in carbon have shown the necessity for further study of such solutions.

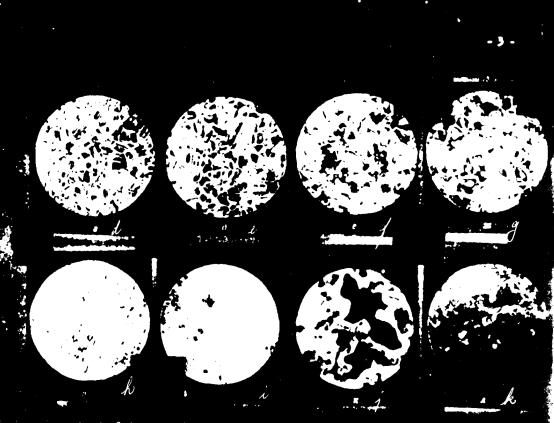
Baring the proparation of boron-carbon solid solutions, the authors used highly refined emorphous boron of 99.7% purity and highly refined lemphlack containing 99.8% earten. These pure boron and earten samples were mixed in desired ratios so that the following compositions were obtained as empressed in terms of carbon 5, 8, 10, 15, 15, 18, 20, 30, to, 70, 60, 70 male \$ (5.6, 6.6, 10.98, 14.14, 16.30, 19.79, 21.75, 32, ta.4, 72.2, 62.2, 72.5 weight \$). These mixtures were then sintered and hot-pressed, using the method employees to the one described by Glaser?. It was found that the malt contacting of 5 male \$ carbon in boron had a lover multing point than that of pure boson (2075°0). The chemical analysis of resulting malten compositions have shown the change in their carbon context so that the actual determined composition of the resulting malts were: 5.11; 9.95; 12.71; 15.89; 17.47; 22.85; 24.87; 35.97; h2.70; 72.60; 62.70; 72.50 weight \$ carbon. The obtained sintered samples of the above boron-carbon compositions were further treated in a vacuum furnace in order to eliminate internal strains and to incure homogeneity of each sample. This treatment was fullowed by also cooling of each sample to rean temperature. Samples for metallographic analysis were prepared from the courses by a diagonal out of each vacuum heat treated sample by the mathed described by Thursdow's.







Figure 1. Microstructure of the boron-carbon alloy samples (magnification (M500)) Alloy compositions in \S C. a - 5.11 \S before heat treatment, b - 5.11 \S after heat treatment, c - 9.93 \S .



d = 15.89%, e = 17.47%, f = 22.38%, g = 24.07%, h = 33.97%, i = **42.50%** j = 52.60%, k = 62.50%.

In figure 1 we show the structure of certain samples of boron-carbon alloys. The sample containing 5.11% C before heat treatment enhibits a polyhedral structure while after heat treatment one notices the supervises of needle like crystals presumably denoting a sutectic alloy. With the rise in the carbon content of the boron carbon samples 9.9% C the supervision in the carbon content of the boron carbon sample area, tends to lesse the needle like crystalline appearance and becomes polyhedral in its character. It is necessary to note that the structure of the alloys containing from 9.95-24.07 weight § carbon is similar before and after the vacuum heat treatment. Starting with 12.71 weight § carbon one notices a certain amount of emplication somewhat diminishing only in a sample containing 24.0%C. However a portion of the above mentioned crystalline sample does not exhibit deplication and has a bright surface which presumably testifies to the presume of two types of crystalline species. In the micrograph of the sample containing 35.9%C a crystalline entectic is noted. (see figure 1 h) Upon further increase of the carbon content of boron-carbon alloy samples a new phase appears on the sutectic background. The maximum amount of this phase was noted in a sample containing (2.5%C. All the samples, after photonicrographic enalysis, were crushed to a fine jowier and then subjected to the x-ray analysis which were carried out v.th copper source and nickel filter. The results of the x-ray analysis are shown shapplically in figure 2.

14/12

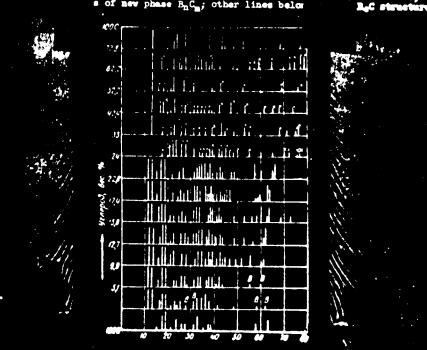


Figure 2. X-ray diagrams of B-C systems; C is carbon lines; R Boron lines; s of new phase $B_{\rm B}C_{\rm B}$; other lines below ReC structure

The study of the x-ray diagram shows that even t . m-carbon melt containing as little as 5.115C exhibits a different pattern is liceting a crystalline syructure different from pure boron. The structure of the main phase of this sail may be described as a structure of a new boron-carbide (B₁₀C₀ or B₁₀C₀).

De the sample containing 5.11\$C along with the lines pertaining to berom carbide (\$A_C\$) one also notes lines of pure boron which in conjunction with the data obtained from microphotographic analysis allows to postulate the presence of solid solution of carbon in boron slong with the conventional beron earlies (\$A_C\$). / Up until the 22.85 we.\$A_C\$ carbon the samples of nelts contain only the \$A_C\$ lines. With the sample containing 24.07 weight \$ carbon the x-ray diagram character is changed. However the characteristic doublet of boron curicles change in the structure of the alloys. Along with the lines of free change in the structure of the alloys. Along with the lines of free change in the appearance of a number of new lines which are most new the x-ray diagram of the melt containing 72.5\$C.

For determination of specific gravities of powdered samples of the becomcarbon melts picnometric method of analysis was used with xylene. The resulting date, listed in Table 2 and see printed in Figure 3. It can be seen from these date that the specific gravity of the boron-carbon melts increased to the maximum 20% earbon after which it begins to decline. These data therefore conflict the date of Glaser?. The micronardness of melts, measured with the aid of the PGF-3 apparatus using a 40 gram load (see table 5 and figure 3) also increases as a function of carbon contents (from 5.11-17 475C) after which it drops repidly. The microhardness of the main phase in a system containing 42.55C is of the same order of magnitude as that of the boron-carbide containing 17.475C.

Carbon Content Weight∮	Density gm/cm	TIES OF BOROS-CARBOS Microhardness Kg/mm	Electric Resistance obs/cm
5.11	2.310	2995+443	1.64 . 10 1
9.93	2.400	3042+497	2104 . 20 -
12.71	2.458	5580 + 501	4.95 7 10-1
15.89	2.531	4460-843	5.14
17.47	2.548	31087463	J.14
22 .83	2.456	4252 F427	4.36
24.07	2.503	2790+117	7.15
33.97 42.5	2.37		1.27
	2.31	-	1.7
52 .6	2.47	44307435	0.8
62.5	2.10	7-2-77	0.0
72.5	2. 0 8	-	-
		_	

Electrical resistance of the boron carbide samples containing 5.11% carton falls rapidly in comparison with the electrical resistance of pure boron after which it increases are in a maximum in the melt containing \$0.000 content.

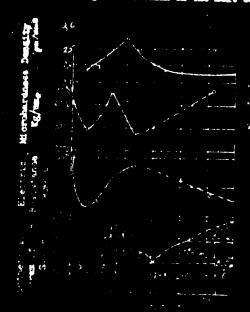


Fig. 3. Electrical resistance, density and micronardness of boron carbon alloy samples as a fair time of partial content

- 6 -

The character of the change in the electrical resistance of the boron carbon alloy samples with the increase in their carbon content allows to postulate following hypothesis. The electrical resistance of boron is in the order of 7.7° x 10° obm/cm and which may be accounted for, supposedly, by the complete compensation of all the valence bonds of boron atoms in the boron crystal. Upon addition of certain amounts of carbon to boron there occurs a change in the structure of the crystalline unit cell which however, leaves some unsatisfied -C-C- bonds which would account for appearance of a certain amount of electrical conductivity. With further addition in the carbon content of some of the boron carbon alloys these unsatisfied -C-C- bonds are compensated and form conventional covalent bonds, common to the conventional boron carbide crystal, this in turn of course leads to the increase in the electrical resistance. The maximum resistance corresponds to the completely satisfied boron carbon matrix with the conventional, directional, covalent bonds of the boron carbon matrix with the conventional, directional, covalent bonds of the boron carbon boron carbon ca

Upon addition of carbon to boron in the amount of 2-3% a sutectic is formed containing boron or a solid solution of carbon in boron with boron carbide B₁₃C₂, which contains as the results obtained from determination of microhardness, density and electrical resistance, an imperfect crystalline structure, that is, vacant spaces in the -C-B-C- of the boron carbide B₁₃C₂.

Upon further increase in the carbon content of the boron-carbon alloys there occurs a gradual "filling-in" of the vacancies in the crystal shell until the carbon content reaches the level equivalent of the carbon content in $B_{18}C_8$. Boron carbide $B_{18}C_9$ is in turn capable of forming solid solutions with carbon eventually forming the $B_{18}C_3$ carbide (that is, forming -C-C-C- in addition to -C-B-C- bonds). This latter carbide forms eutectic compositions with the carbide $B_{18}C_{18}$, which is richer in carbon than B_4C and which has a tentative formula BC_2 . This carbon rich boron carbide material (eutectic) is the product of the following reaction, $B_{18}C_{18} \rightarrow B_4C+C$, which is reversible and is usually accompanied by formation of free carbon. On the basis of the above hypothesis we have constructed a phase diagram describing the boron carbon system containing up to 600-70% carbon shown in Figure 3. It has to be noted that this system satisfactorily agrees with the data described in earlier works pertaining to the study of boron carbides.

CONCLUSIONS

1. An x-ray and a photomicrographic study of the boron carbon system has been made which includes determinations of electrical resistance and densities of the boron-carbon melts in the region of carbon content from 5 to 72.5 weight percent carbon. In this study we show that with the 5% carbon and higher content in the boron-carbon melts there is produced a defective crystalline structure of boron carbids $B_{13}C_2$ which is characterized by the presence of vacancies in the -C-C-calong the crystal axis of $B_{13}C_2$. In the 14-16% carbon containing melts these vacancies are filled with the additional carbon atoms forming the -C-B-C-structure, which is then further changed to the -C-C-C-structure in the $B_{12}C_2$ (B4C) boron carbide. The melts containing the defective crystalline structure of $B_{13}C_2$ possess a considerable amount of electric conductivity which is dependent on the amount of vacancies occurring along the C axis.

- 7 -

2. On the basis of previously obtained data and also results obtained in this present work an attempt was made to depict a hypothetical phase diagram for the boron-carbon system.

LITERATURE

- 1. Ridgway, R. R. Trans. Am. Electrochem. Soc., 1934, 56, 117-133.
 2. Ormont, B. F. Structure of Inorganic Compounds, 1950, 689.
 3. Meerson, et al, I.S.F.H.A. A.N.SSSR, 1953, 22, 92-103.
 4. Zhdanov, et al, J.F. Kh. 1945, 42, 326-335.
 5. Zhdanov, et al, D.A.N. SSSR, 1955, 92, 4, 767-768.
 6. Zhdanov, et al, J.F.Kh, 1954, 28, 1076.
 7. Glaser, F., Moskowitz D. a. Post B Journ. Appl. Phys., 1953, 24, 6, 731-733.
 8. Allen, R., Journ. Am. Chem. Soc., 1953, 72, 3582-83.
 9. Sameonov et al, Strudain Troudov, M.I.Ts. Mi.Z. 1956, 27.
 Filiand, M. A. et al, Properties of Rare Metals, 1954.

Translated by: V.V. Levasheff 2/28/58